

Polarization sensitive solar-blind detector based on *a*-plane AlGaN.

Masihur R. Laskar, A. Arora, A. P. Shah, A. A. Rahman, M. R. Gokhale, Arnab Bhattacharya

^a*Department of Condensed Matter Physics and Materials Science,
Tata Institute of Fundamental Research, Homi Bhabha Road, Mumbai 400005, India.*

Abstract

We report polarization-sensitive solar-blind metal-semiconductor-metal UV photodetectors based on (11̄20) *a*-plane AlGaN. The epilayer shows anisotropic optical properties confirmed by polarization-resolved transmission and photocurrent measurements, in good agreement with band structure calculations.

Solar blind UV (SBUV) detectors, with no photosensitivity above 280nm wavelength, have wide range of applications like – missile plume detection, UV astronomy, chemical/biological battlefield reagent detection etc.^{1–3}. The wide-bandgap, high-temperature compatible AlGaN material system has been the workhorse for such SBUV detectors with many reports on high performance devices based on [0001] *c*-plane AlGaN layers. The inherent anisotropic optical properties and reduced crystal plane symmetry of “non-polar” (11̄20) *a*-plane AlGaN epilayers allows the fabrication of polarization sensitive (PS) detectors. Such PS detectors give additional advantages of selectivity and narrow band detection in a differential configuration consisting of two or four photo-detectors, without using filters^{4,5}. We present, to the best of our knowledge, the first report of a PS SBUV detector.

About $0.5\mu\text{m}$ thick $\text{Al}_{0.6}\text{Ga}_{0.4}\text{N}$ epilayers were grown on AlN buffer layers via metal organic vapour phase epitaxy (MOVPE) in a closed-coupled showerhead reactor using standard precursors. The details of the growth procedure, method to estimate the solid phase Al content and strain in the layer can be found in Refs.[6,7]. Metal-Semiconductor-Metal (MSM) type devices with interdigitated finger geometry Schottky contacts (metallization–200Å Ni/1000Å Au) were fabricated using standard optical photolithography, electron-beam evaporation and lift-off techniques.

The III-nitride semiconductors have three closely-spaced valence bands near the center of

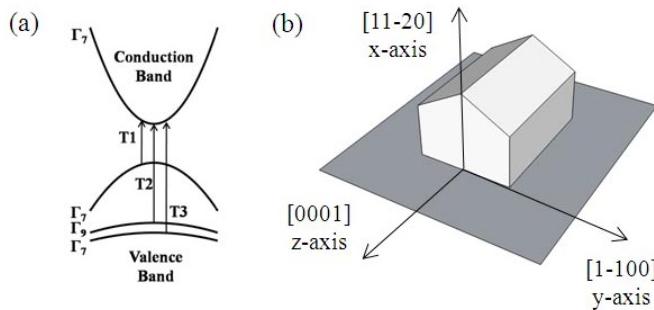


Figure 1: (a) Schematic diagram showing the three closely spaced valence band at $k=0$ of the III-nitrides. (b) Orientation of hexagonal unit cell for *a*-plane nitrides. The in-plane strains are ϵ_{yy} and ϵ_{zz} .

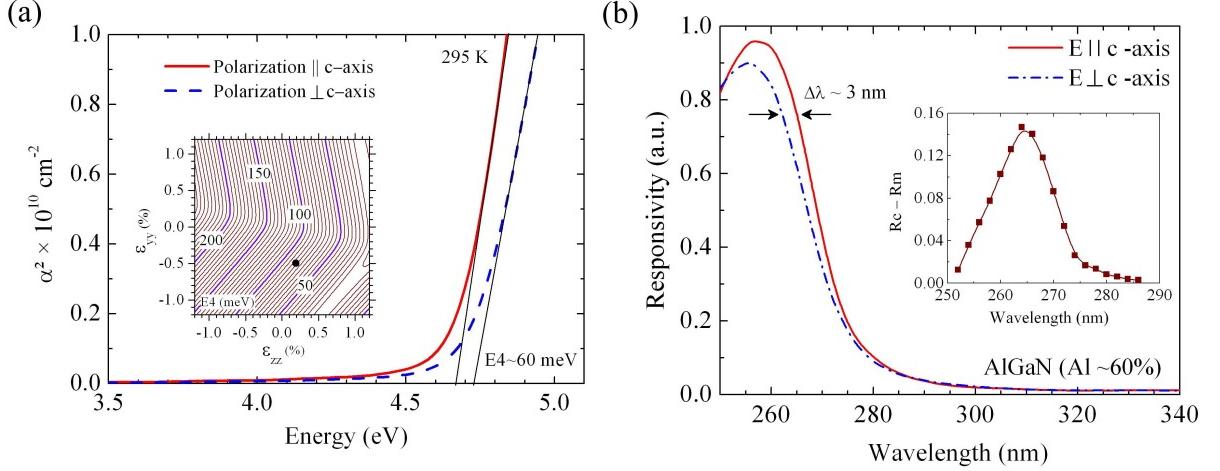


Figure 2: (a) Optical absorption spectra of *a*-plane $\text{Al}_{0.6}\text{Ga}_{0.4}\text{N}$ showing difference in bandgap $E_4 \approx 60$ meV for two different polarizations, Inset: calculated E_4 as a function of in-plane strains, black dot represent the strain in our layer for which the calculated value is ~ 80 meV (b) Polarization resolved photocurrent measurement for $E \parallel c$ and $E \perp c$ polarization, confirming polarization sensitivity with sharp cut-off below 280 nm. Inset: different in responsivity as a function of wavelength.

the Brillouin-zone ($k=0$) as shown in Fig.1(a). The transition probabilities of electrons from each valence band to the conduction band are different and are strongly determined by the polarization of light. For $(11\bar{2}0)$ *a*-plane epilayers, the in-plane strains are ϵ_{yy} and ϵ_{zz} as shown in Fig.1(b). Using HRXRD we estimate the in-plane anisotropic strain in our $\text{Al}_{0.6}\text{Ga}_{0.4}\text{N}$ epilayer as $\epsilon_{yy}=-0.5\%$ and $\epsilon_{zz}=+0.2\%$, for which $E1$ transition is strongly ***z***-polarized and $E2$ transition is strongly ***y***-polarized, obtained from the band structure calculation by solving the *Bir-Pikus* Hamiltonian^{8,9}.

Fig.2(a) shows the absorption spectra of $\text{Al}_{0.6}\text{Ga}_{0.4}\text{N}$ for two different polarizations, where the extrapolation of α^2 vs. *energy* plot gives the bandgaps of the epilayer as ~ 4.67 eV and ~ 4.73 eV for $E \parallel c$ and $E \perp c$ polarization directions respectively. So the valance band splitting $E_4=E2-E1$ is ≈ 60 meV. Fig.2 (a) inset shows the calculated E_4 as a function of in-plane strain and the the black dot represents the strain in the layer. The experimentally obtained value of E_4 fairly matches with the value 80 meV obtained from calculation.

The polarization-resolved photocurrent measurement on the device (geometry: finger width $10\mu\text{m}$ and gap $10\mu\text{m}$; bias voltage=10 V) fabricated on $\text{Al}_{0.6}\text{Ga}_{0.4}\text{N}$ shows different responsivity spectra R_c and R_m for different in-plane polarization $E \parallel c$ and $E \perp c$ respectively, as shown in Fig.2(b). Inset shows the difference in responsivity ($R_c - R_m$) as a function of wavelength. It shows a peak at ~ 265 nm with peak responsivity of $\sim 15\%$ to the maximum responsivity R_c and FWHM of $\sim 10\text{nm}$. The UV to visible rejection ratio is 10^2 . The polarization sensitivity contrast (R_c/R_m) is about 1.2. Both the spectra shows cut-off below 280nm, fulfilling the solar-blind criteria, and making this perhaps the first demonstration polarization sensitive SBUV detectors reported so far.

In conclusion, we have successfully demonstrated polarization-sensitive SBUV detectors fabricated on non-polar *a*-plane AlGaN. Such devices will be helpful for civil and strategic applications.

References:

- [1] E. Monroy *et al.* Semicond. Sci. Technol. **18** (2003) R33-R51.

- [2] M.A. Khan *et al.* Jpn. J. Appl. Phys. **44** (2005) 7191-7206.
- [3] M. Razeghi *et al.* J. Appl. Phys. **79** (1996) 7433.
- [4] S. Ghosh *et al.* Appl. Phys. Lett. **90** (2007) 091110.
- [5] A. Navarro *et al.* Appl. Phys. Lett. **94** (2009) 213512.
- [6] M. R. Laskar, *et al.* Phys. Stat. Sol. (RRL) **4**, (2010) 163.
- [7] M. R. Laskar, *et al.* J. Appl. Phys. **109**, (2011) 013107.
- [8] J. Bhattacharya *et al.* Phys. Status Solidi B **246**, 1184 (2009).
- [9] M. R. Laskar, *et al.* Appl. Phys. Lett. **98**, (2011) 181108.